

УДК 544.18+538.955:549.211-022.532

Nanodiamond Collective Electron States and their Localization

Ivan A. Denisov*

Andrey A. Zimin†

MOLPIT, Institute of Fundamental Biology and Biotechnology,
Siberian Federal University,
Svobodny, 79, Krasnoyarsk, 660041
Russia

Leslie A. Bursill‡

School of Physics, The University of Melbourne,
PO BOX 339, Carlton North VIC 3054,
Australia

Peter I. Belobrov§

MOLPIT, Institute of Fundamental Biology and Biotechnology,
Siberian Federal University,
Svobodny, 79, Krasnoyarsk, 660041
Kirensky Institute of Physics & Institute of Biophysics SB RAS
Akademgorodok, 50/50, Krasnoyarsk, 660036
Russia

Received 15.11.2013, received in revised form 22.11.2013, accepted 22.11.2013

The collective states despite their importance are rarely used to describe the electron structure of dielectric materials. The nature of the unrelated to impurities unpaired spins found experimentally in the nanodiamond is still under discussion. We propose the explanation of their nature in terms of the collective electron states. Collective states are studied by solving a one-particle one-dimensional Schrödinger equation in the Kronig–Penney potential and by ab initio computations of ground state wavefunctions of diamondoids $C_{78}H_{64}$, $C_{123}H_{100}$ and $C_{211}H_{140}$ at the DFT R-B3LYP/6-31G(d,p) level of theory. Three distinct classes were found: collective bonding states resembling modulated particle in a box solutions; surface-localized non-bonding conductive Tamm states and subsurface-localized bonding states for non-uniformly compressed nanodiamond. The existence of the unpaired spins is supposed to result from the spin-density fluctuation effects significant on the nanoscale collective and spread subsurface states.

Keywords: nanodiamond, nanoscale, collective electron states, Tamm states, subsurface electron states, nanodiamond paramagnetism, spin-density fluctuations.

Introduction

Collective electron states are widely used in the theory of metals [1], while bulk dielectric and semiconductor materials are equally well modeled as a rigid network of covalent bonds incorporating lattice defects. Success for classic models is mainly due to extensive use of translational symmetry, which is definitely not the case for nano-sized systems: nanoparticles are mostly imperfections with perfect regions rather than perfect crystals with imperfections. Lattice termination is the main imperfection leading to the consideration of the collective surface-localized

*d.ivan.krsk@gmail.com

†andrey.alx.zimin@gmail.com

‡lesbursill@hotmail.com

§peter.belobrov@gmail.com

electron states problem. However, one should clearly distinguish collective states in a metal and in a dielectric material — collectivity doesn't necessarily imply conductivity, and there is no contradiction between the “bonding” and “collective” terms. It's crucial to emphasize unambiguous meaning of the term “collective” we use in our paper: collective states in dielectric materials are bonding molecular orbitals, localized over several atomic cores. The fact that collective states despite their importance are rarely used in dielectric materials theory results in seemingly unsolvable issues like inability to describe the nature of the intrinsic spin found experimentally in the nanodiamond.

B. Pate has summarized study of the electronic and atomic structure of the bulk diamond in his classic paper [2]. Almost simultaneously the nanodiamond was discovered, and its practical applications attracted great attention [3, 4], leaving behind detailed investigation of the underlying structural and electronic aspects. Nanodiamond-based materials are rapidly growing in number, triggering progress in the understanding of the CVD synthesis methods [5] and in the thermophysical applications of nanodiamond-based materials [6]. Despite great success in nanodiamond applications, there is still no in-depth understanding of its structure. Collective electron states are vital part of the global perspective which interrelates consistently magnetism, surface conductivity and the set of nanodiamond experimental “anomalies”.

The first theoretical investigation of the collective electron states in a bulk diamond was done by I. E. Tamm [7]. His calculations were based on the idea of electron confinement between periodic lattice potential and vacuum. We have followed Tamm approach assuming that electron could be confined between vacuum and semi-periodic diamond ball kernel.

J. Koutecký then investigated surface states of a one-dimensional crystal in connection with penetration of surface potential perturbation into the crystal and was the first to introduce the term “subsurface states” [8]. P. Phariseau then studied energy spectrum of amorphous substances and subsurface states in deformed one-dimensional crystals, treating deformed region as amorphous phase [9]. Direct experimental evidence of the surface layer deformation in nanodiamond molecules follows from the Parallel EELS (PEELS) [10, 11], NMR [12] and Auger [13] data and allows us to make the assumption about subsurface states existence in nanodiamond.

We have concentrated our attempts on study of the collective electron states in nanodiamond in order to relate great amount of well-documented reliable experimental data with the *ab initio* computations of its electronic structure. It was recently demonstrated that it's vital to take nuclear movement explicitly into account in order to describe optical gap of diamondoids [14]. However, full-scale *ab initio* geometry optimization at meaningful level of theory is still impossible for big diamondoids. The optimization procedure must be consistent with the experimentally observed deformation. We introduce an approximate parametric non-uniform compression model in order to skip the geometry optimization of the diamond ball and to catch the exaggerated influence of deformation on small diamond ball models.

The problem of collective electron states localization in nanodiamond is considered, and qualitative explanation of the nature of the experimentally observed compression, the intrinsic unpaired spin and the PEELS pre-peak is proposed.

1. Methods of electronic structure investigation

Orbital localization in nanodiamond was studied on model systems by exact numerical integration of Schrödinger equation for one-particle model in one-dimensional Kronig–Penney potential and by *ab initio* computations of ground-state wavefunctions of small diamondoids. Simultaneous usage of both methods allows to establish their ranges of applicability and to give better approximation to real 4–6 nm nanodiamonds.

1.1. 1D collective electron states analysis

A one-dimensional one-electron model allows to analyze wavefunctions of big diamond balls (5 nm and bigger). This model gives clear outcome and its solutions are exact. Despite relative simplicity of one-electron approximations it is widely used in classical solid state physics. Our calculations are made following the logic of I. E. Tamm [7] and A. A. Abrikosov [1] using the test electron approach and the model of the Kronig–Penney for one-dimensional dielectric crystal, bounded on both sides by vacuum. We consider intermediate case between pure dielectric potential in tight-binding approximation and metallic lattice potential where “free” electron gas moves in atomic core potential.

Special application was developed to explore the one-dimensional stationary Schrödinger equation solutions in a limited quasi-periodic potential using Component Pascal programming language in the BlackBox Component Builder (Oberon microsystems AG, Zürich, Switzerland).

1.2. Design of 3D models of diamond balls

A set of diamond ball structures in a format compatible with a quantum-chemical software is necessary before one can perform numerical computations of wavefunctions and energies in the three-dimensional case. Application DiaBall (<http://diaball.molpit.com>) was developed to construct diamond balls up to 10 nm size, visualize and export molecular model in the PDB format. The value of the lattice constant is taken to be 0.35669 nm [15] in order to build carbon diamond lattice inside defined radius. Two different unit cell arrangements (atom-centered or volume-centered) are possible, varying diamond ball structure slightly.

The structure of the compressed layer is usually and erroneously understood in terms of sp^2 -shell coating diamond core, despite the vast amount of experimental data showing inadequacy of this model. Thus, NMR spectroscopy is unable to detect a sp^2 phase signal [12] above the detection limit; neither can X-ray powder diffraction. We conclude that the compressed layer is not to do with sp^2 -phase necessarily, so every atom in our model (and in real best quality nanodiamond, actually) behaves as if it has same coordination as sp^3 -hybridized carbon.

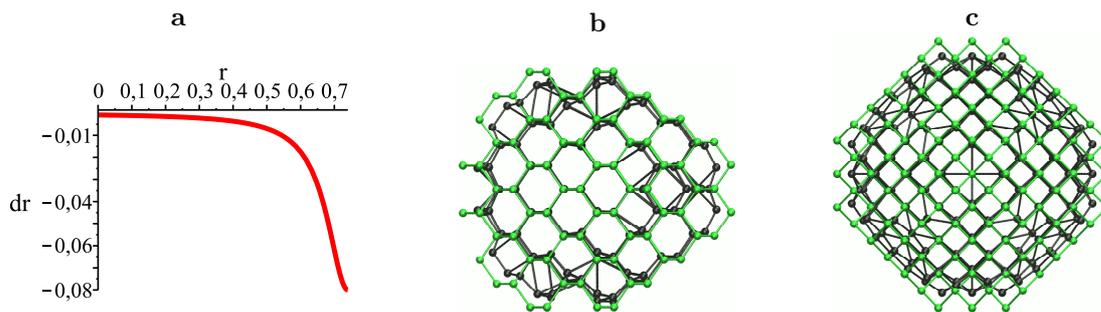


Fig. 1. (a) Magnitude of the atom position shift to the origin vs. coordinate (fitted to the function (1) with parameters $s = 10$, $a = 0.08$). (b, c) Compression of the C_{302} diamond ball according to the function (1): initial diamond ball (green) and deformed (black)

Software realization of controlled compression of built structures is shown at Fig. 1 and is characterized by parameters a and s according to the next function:

$$dr(r) = -\frac{a}{(s - rs/R)^2 + 1}, \quad (1)$$

where dr is the atom position shift relative to the origin, r is the distance from atom to the origin, R is the radius of molecule, a is the parameter regulating amplitude of shift and s is the parameter regulating shift decay rate inside diamond ball.

The chosen compression method (1) allows roughly to estimate the relaxed diamond ball shell structure under surface forces resulting from self-consistency of the electron-nuclear interaction. A quantitatively correct description of the compression is much more complex and, more importantly, would provide higher precision than we really need for demonstration purposes.

1.3. *Ab initio* diamondoids electronic structure computations

Ground-state electronic structure computations for several small diamondoids $C_{78}H_{64}$, $C_{123}H_{100}$ and $C_{211}H_{140}$ were performed at DFT R-B3LYP level of theory using the 6-31G(d,p) basis set. Three cases were considered: non-deformed particle and two deformed configurations with compression parameters $s = 10$, $a = 0.04$ and $s = 10$, $a = 0.08$ applied according to eq. (1). Computations were made in the GAMESS-US [16] software package (May 1, 2012 R2 version). Several pure RHF computations were performed with the same basis set and have shown that wavefunction morphology is preserved in both RHF and DFT computations, thus allowing one to use Kohn–Sham (KS) orbitals as a qualitatively correct basis for molecular orbital picture interpretation. Strictly speaking, KS orbitals “are physically sound and may be expected to be more suitable for use in qualitative molecular orbital theory than either Hartree–Fock or semiempirical orbitals” [17], especially in the case of occupied states [18].

The structure of hydrogenated diamondoids was optimized using OpenBabel [19] in the Avogadro [20] software package and MMFF94 [21] force field. Carbon positions were fixed in case of compressed diamondoids and only the hydrogen shell was relaxed.

2. Quantum-mechanical analysis of model diamond balls

We have found that every bonding state in diamond balls is intrinsically collective and we classify them according to three distinct types: bonding orbitals morphologically similar to hydrogenic atom wavefunctions, Tamm surface electron states and subsurface-localized electron states. The results are illustrated both by a 1D one-particle example and by the corresponding *ab initio* computations.

It is worth noting that bonding molecular orbitals for 1D one-particle solutions and for spherical diamondoids (Fig. 2) give solutions with shapes and nodal structure similar to modulated hydrogenic atom wavefunctions of s , p , d shape, etc. Generally, the observed orbital morphologies arise from near-spherical symmetry of the system and similarity with the particle-in-a-box case. We should mention that collective nature of the molecular wavefunction is not a striking new property: Dewar [22] had pointed out in 1984 that “bond localization is not a real phenomenon” and description of molecules in terms of localized bonds is wildly incorrect. Here we emphasize his thoughts and results as applied to the description of nanodiamond electronic structure.

As shown in Fig. 3, the electron density of some quantum states (bold lines on the plot) is located near the surface layer and their energy levels are located between the valence and conduction bands. The total number of states in valence and surface bands are equal to the number of potential wells. In accordance with earlier discussed classical theoretical results surface state penetration into the crystal is observed if potential holes are shallow enough. The energy level of the surface states is almost parallel in the plot of energy (Fig. 3) versus lattice constant, and can be either below or above the Fermi energy (denoted as zero line).

Consideration of the 3D problem allows to investigate spatial localization of Tamm states. It’s unclear from 1D calculations if Tamm states are local in a sense of dangling bond localization or have intrinsically collective nature. It’s still impossible to compute the optimized electronic struc-

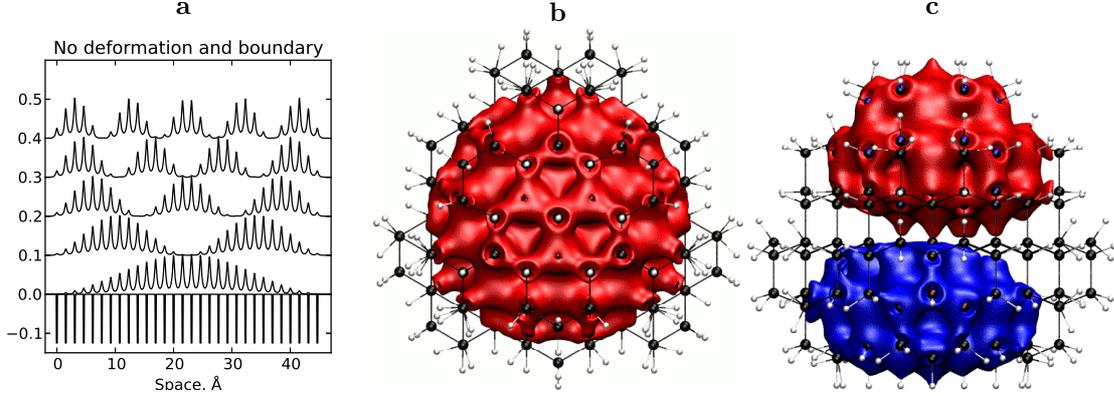


Fig. 2. (a) First five bonding molecular orbitals for 1D case are resembling modulated particle-in-a-box solutions. (b, c) Isosurface of the first two bonding molecular orbitals #212 and #213 of $C_{211}H_{140}$ molecule resemble typical shapes and nodal structure of atomic s and p orbitals. Visualization is made in the VMD [23] software package using PovRay (<http://www.povray.org/>)

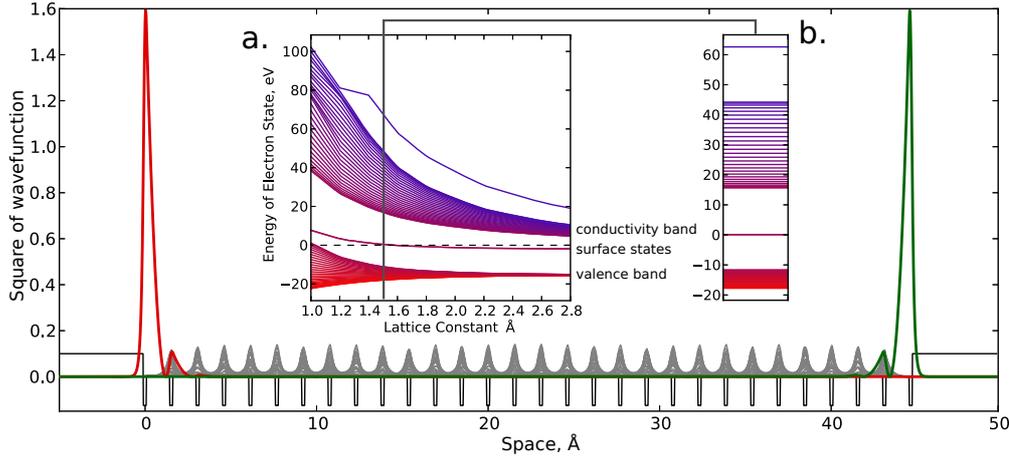


Fig. 3. Square of electron wavefunctions in limited periodic potential (holes deepness 25 eV, width 0.2 Å, lattice constant 1.54 Å, boundary $5 \text{ \AA} \times 20 \text{ eV}$). (a) Energy spectrum versus lattice constant of one-dimensional crystal. (b) Energy spectrum for 1.54 Å lattice constant

ture of real 5 nm diamond ball in the three dimensional case. That is why we have investigated surface states of computationally feasible 1.34 nm diamondoid $C_{211}H_{140}$. Tamm states in the 3D singlet case are located inside virtual orbitals band and show collective and surface-localized wavefunction nature. The HOMO and LUMO states for the compressed and non-compressed diamondoid $C_{211}H_{140}$ are shown on Fig. 4.

Assuming that 70% of diamond ball sp^3 -bonds are deformed [12], the radius of the uncompressed core of the diamond ball approximately equals to 89% of the ball radius.

As can be seen from Fig. 5, a surface Tamm states are localized on both sides of crystal. Symmetrical shift of the edge potential holes according to (1) results in subsurface localization of wavefunctions (Fig. 5, b). Localization change is observed in the lowest and highest (nearest to the surface states) wavefunctions. Subsurface states show a symmetrical split in the case of

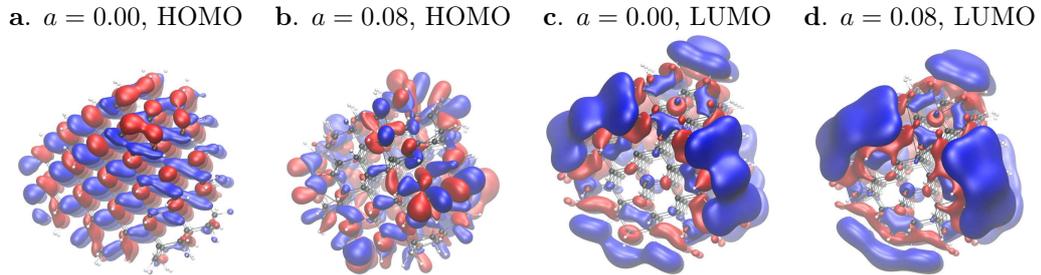


Fig. 4. Isosurfaces of wavefunction of 1.34 nm diamondoid $C_{211}H_{140}$ (red 0.01 a.u., blue -0.01 a.u.)

relatively low compression, becoming asymmetrical in the presence of high compression and show significant splitting from the valence band (Fig. 5, b).

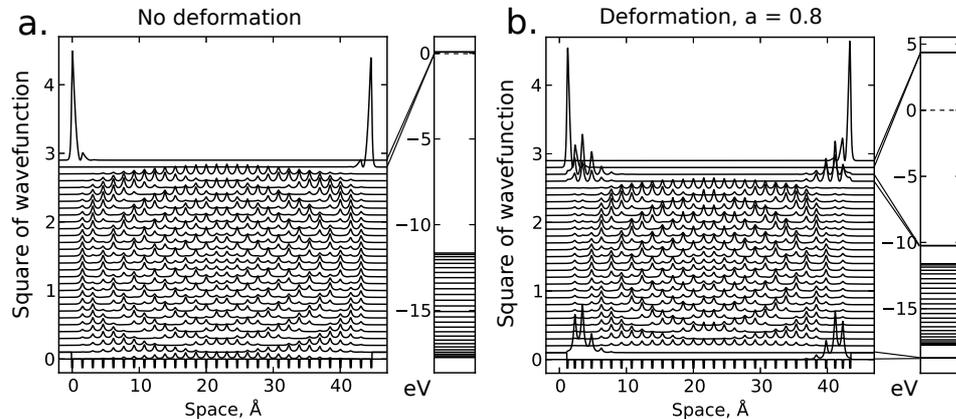


Fig. 5. Collective 1D electron states for: (a) Tamm surface states with barrier, (b) subsurface states in presence of barrier and compression

The lowest bonding molecular orbital morphology for uncompressed, slightly compressed and heavily compressed cases is shown in Fig. 6. Nine diamondoids are considered. The general tendency could be seen in all cases, corresponding to orbital localization flowing into the subsurface area. Consideration of the $C_{78}H_{64}$ diamondoid (Fig. 6, a–c) shows slightly pronounced shift of the first bonding molecular orbital to the subsurface area, while bigger diamondoids $C_{123}H_{100}$ (Fig. 6, d–f) and $C_{211}H_{140}$ (Fig. 6, g–i) show dramatic orbital localization shift. Another interesting point is that in uncompressed diamondoids (Fig 6, a, d, g) first bonding molecular orbitals is almost identical.

3. Discussion

Examination of our computations shows the existence of three distinct types of solutions for both 1D and *ab initio* 3D case. We provide a qualitative interpretation of the nature of nanodiamond “disordered” shell, the EPR signal invariant and the possible source of the unusual pre-peak in PEELS experiment. Explanation is provided in terms of collective electron state localization in nanodiamonds.

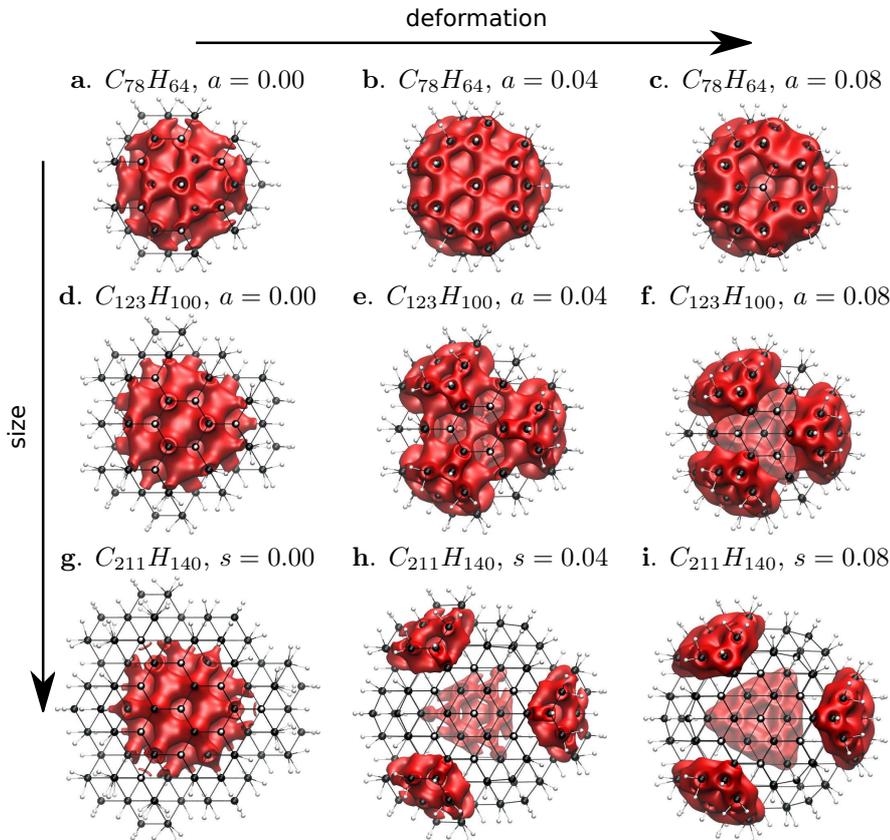


Fig. 6. Wavefunctions isosurfaces (0.02 a.u.) for the lowest bonding orbital of diamond balls of three sizes: (a–c) $C_{78}H_{64}$, (d–f) $C_{123}H_{100}$, (g–i) $C_{211}H_{140}$ and three fixed compressions

3.1. Tamm surface states

According to Tamm reasoning for a ball-shaped dielectric nanocrystal electron can be confined in a spherical layer between the vacuum and the periodical crystal potential. The surface electron band lies between the valence band and the conduction band. Wavefunctions of the surface states are localized near the surface and form a spherical spatial layer if we extrapolate one-dimensional case to three dimensions. Tamm electrons are locked in the radial degree of freedom but free in the angular degree of freedom, moving on the surface and belong to the whole particle as if it is a single sheet. Surface localization of wavefunctions give few solutions localized on the nanodiamond; rather they float as the surface electrons.

Such “floating electrons” in a bulk diamond are discussed as alternative to classical electrical contacts or leads in new electronic devices [24] and become the base for diamond electronic devices. As shown in experiments, surface conductivity is sensible to modification of surface, and its existence demands hydrogen terminated surface [25]. However, unpaired electrons lie under the surface of nanocrystal according to NMR data [12], and EPR properties are also independent on surface modification [26]. That is why Tamm surface states can not explain paramagnetic properties of nanodiamond [27]. Nevertheless, we suggest to take into account surface states for future exploration of the transport properties of composite materials based on nanodiamond [28].

Energy localization of these states in the middle of the band gap could be easily explained

if we realize that one-particle one-dimensional model represents one-dimensional doublet radical state. Semiempirical simulations of n -mantane ($C_{60}H_{60}$) radical electron structure [13] show direct correspondence with the one-dimensional case, where radical energy level lie in the middle of the band gap. If we consider singlet case, Tamm states would be located in the virtual orbitals band.

3.2. Subsurface states

PEELS of single nanodiamond [11, 13] (the ratio of the pre-peak to diamond peak intensity) shows that electron structure of the surface differs from the electron structure of the diamond core. Auger electron spectroscopy [13] shows clearly distinguished spectrum identifying unusual chemical states of electrons in nanodiamond. Evidence for core-shell structure of nanodiamond is presented in NMR study [12] indicating 39% atoms in core, 40% of partially disordered five-layered shell and remaining 21% attributed to two nearest to surface layers.

Strictly speaking, deformation is represented by surface layer compression which is the natural consequence of electron and nuclear structure self-consistency. Recent large-scale *ab initio* geometry optimization of diamondoids up to 2 nm size at DFT/6-31G within LDA level of theory shows a gradual increase in subsurface-localized character of HOMO as particles grow [29]. It means that if one perform “honest” full-scale geometry optimization at an acceptable theory level subsurface states arise naturally. However, optimization of these structures is extremely demanding and time consuming.

We conclude that the rough compression model we employed allows qualitatively correct results to be obtained concerning orbital localization without employing full-scale first-principles optimization procedures.

To sum up, artificial compression gives similar subsurface states as first-principles geometrical optimization. It’s argued that first-principles optimization naturally results in particle non-uniform compression leading to the subsurface localization of bonding orbitals. More importantly, these subsurface-localized bonding orbitals are collective, i.e. spread over large amount of carbon atoms located in appropriate shell.

Considering our results and large-scale computations done by Jiang et al. [29] we extrapolate subsurface orbital model to real-sized 5 nm nanodiamond. It appears then that these orbitals have large spatial extent, allowing us to use analogy with the long conjugated π -electron systems and providing an opportunity to discuss the nature of intrinsic spin in nanodiamond.

3.3. Intrinsic spin nature

Nanodiamond exhibits unusual spin states in wide range of experiments, including EPR spectroscopy [26] and NMR relaxation [12, 31]. It demonstrates uncommon to the bulk phase paramagnetic behavior. Observed unpaired electron behavior in nanodiamond couldn’t be explained in terms of quite typical localized radical state, e.g. as an F -center, because it will necessarily yield characteristic change in optical properties of the system which is not observed experimentally. If radical state is not localized on particular defect, it must be delocalized in some sense.

Taking into account subsurface localization of these unpaired spins, we attribute them to collective subsurface orbitals. These subsurface orbitals have comparatively long extent, and spin-density fluctuations could possibly explain existence of intrinsic unpaired spin in nanodiamond. Analogy may be made with long conjugated π -electron systems where singlet instabilities of ground-state solutions are widely known [32]. Fluctuational nature of intrinsic spin is probably the only suitable explanation which does not contradict experimental data, chemical reason, charge or high-spin states issues. There is no way to imagine classical radical spin-1/2 state localized in the subsurface layer and being nevertheless stable. Moreover, it seems that the

number of intrinsic spins depends on the magnitude of applied field which probably magnifies the spin-density fluctuations in the subsurface orbitals. Fluctuational mechanism of “unpaired” electron existence is consistent with the fact that nanodiamond has no unpaired electrons in absence of external field.

Conclusion

Three distinct classes of collective electron states have been found: collective bonding orbitals resembling the morphology of 3D-modulated particle in a box solutions; surface-localized non-bonding conductive Tamm states and subsurface-localized bonding states for non-uniformly compressed nanodiamond.

First-principles computations unambiguously show that every molecular orbital responsible for chemical bonding is collective. Surface compression plays crucial role in determining orbital localization inside nanodiamond, and we suggest to use “Buried Layer Electron States” term for these states.

On the base of buried layer electron states model we suggest that collective unpaired electrons are intrinsic to nanodiamond. Their subsurface localization is described in terms of surface compression arising from self-consistency of electrons and nuclei positions. Intrinsic spin existence is supposed to result from collective and spread nature of subsurface orbitals, allowing spin-density fluctuation effects to become significant on this length scale. Suggested model could explain free spins exhibited in experiments avoiding contradictions such as when one tries to attribute unpaired or free spin to radicals localized on nanodiamond surface.

Thanks to the Genomic Research and Educational Center of SibFU for the provided access to the supercomputer cluster. The research was supported by the Program of the RF Government “Measures to attract leading scientists to Russian educational institutions” (grant no. 11.G34.31.058) and by the Ministry of Education and Science (Federal Government Contract no. 14.513.11.0123).

References

- [1] A.A.Abrikosov, Introduction to the theory of normal metals, Academic Press, New York and London, 1972.
- [2] B.B.Pate, The diamond surface: atomic and electronic structure, *Surface science*, **165**(1986), 83–142.
- [3] W.Yang *et al.*, DNA-modified nanocrystalline diamond thin-films as stable, biologically active substrates, *Nature Materials*, **1**(2002), 253–257.
- [4] J.E.Butler, A.V.Sumant, The CVD of nanodiamond materials, *Chemical Vapor Deposition*, **14**(2008), 145–160.
- [5] J.E.Butler, Y.A.Mankelevich, A.Cheesman, J.Ma, M.N.R.Ashfold, Understanding the chemical vapor deposition of diamond: recent progress, *Journal of Physics: Condensed Matter*, **21**(2009), 364201.
- [6] A.A.Balandin, Thermal properties of graphene and nanostructured carbon materials, *Nature Materials*, **10**(2011), 569–581.
- [7] I.E.Tamm, Über eine mögliche art der elektronenbindung an kristalloberflächen, *Z. Phys. Sowjetunion*, **1**(1932), 733–746.

- [8] J.Koutecký, Contribution to the theory of the surface electronic states in the One-Electron approximation, *Physical Review*, **108**(1957), 13–18.
- [9] P.Phariseau, The energy spectrum of an amorphous substance, *Physica*, **26**(1960), 1185–1191.
- [10] L.A.Bursill, A.L.Fullerton, L.N.Bourgeois, Size and surface structure of diamond nanocrystals, *International Journal of Modern Physics B*, **15**(2001), 4087–4102.
- [11] J.Peng, S.Bulcock, P.Belobrov, L.Bursill, Surface bonding states of nano-crystalline diamond balls, *International Journal of Modern Physics B*, **15**(2001), 4071–4086.
- [12] X.Fang, J.Mao, E.M.Levin, K.Schmidt-Rohr, Nonaromatic Core-Shell structure of nanodiamond from Solid-State NMR spectroscopy, *Journal of the American Chemical Society*, **131**(2009), 1426–1435.
- [13] P.I.Belobrov, L.A.Bursill, K.I.Maslakov, A.P.Dementjev, Electron spectroscopy of nanodiamond surface states, *Applied Surface Science*, **215**(2003), 169–177.
- [14] C.E.Patrick, F.Giustino, Quantum nuclear dynamics in the photophysics of diamondoids, *Nature Communications*, **4**(2013).
- [15] D.R.Lide *et al.*, Handbook of Chemistry and Physics, vol. 131 (National Institute of Standards and Technology, 2009), 90 edn.
- [16] M.W.Schmidt *et al.*, General atomic and molecular electronic structure system, *Journal of Computational Chemistry*, **14**(1993), 1347–1363.
- [17] E.J.Baerends, O.V.Gritsenko, A quantum chemical view of density functional theory, *The Journal of Physical Chemistry A*, **101**(1997).
- [18] R.Stowasser, R.Hoffmann, What do the Kohn–Sham orbitals and eigenvalues mean?, *Journal of the American Chemical Society*, **121**(1999), 3414–3420.
- [19] N.M.O’Boyle *et al.*, Open babel: An open chemical toolbox., *Journal of cheminformatics*, **3**(2011), 33–33.
- [20] M.D.Hanwell *et al.*, Avogadro: an advanced semantic chemical editor, visualization, and analysis platform, *Journal of Cheminformatics*, **4**(2012), 17.
- [21] T.Halgren, R.Nachbar, Merck molecular force field. IV. conformational energies and geometries for MMFF94, *Journal of Computational Chemistry*, **17**(1996), 587–615.
- [22] M.J.Dewar, Chemical implications of σ conjugation, *Journal of the American Chemical Society*, **106**(1984), 669–682.
- [23] W.Humphrey, A.Dalke, K.Schulten, VMD: visual molecular dynamics, *Journal of Molecular Graphics*, **14**(1996), 33–38.
- [24] M.P.Ray *et al.*, Towards electrons floating over diamond, In Bulletin of the American Physical Society, **56**(2011), 27015.
- [25] F.Maier, M.Riedel, B.Mantel, J.Ristein, L.Ley, Origin of surface conductivity in diamond, *Physical review letters*, **85**(2000), 3472–5.
- [26] P.I.Belobrov, S.K.Gordeev, E.A.Petrakovskaya, O.V.Falaleev, Paramagnetic properties of nanodiamond, *Doklady Physics*, **46**(2001), 459–462.

- [27] E. Levin *et al.*, Magnetization and ^{13}C NMR spin-lattice relaxation of nanodiamond powder, *Physical Review B*, **77**(2008), 1–10.
- [28] S.K. Gordeev *et al.*, Specific features in the change of electrical resistivity of carbon nanocomposites based on nanodiamonds under neutron irradiation, *Physics of the Solid State*, **55**(2013), 1480–1486.
- [29] J. Jiang *et al.*, Structure dependent quantum confinement effect in hydrogen-terminated nanodiamond clusters, *Journal of Applied Physics*, **108**(2010), 094303–094303.
- [30] L. Tian *et al.*, Carbon-Centered free radicals in particulate matter emissions from wood and coal combustion, *Energy & fuels : an American Chemical Society journal*, **23**(2009), 2523–2526.
- [31] A.M. Panich, Nuclear magnetic resonance studies of nanodiamonds, *Critical Reviews in Solid State and Materials Sciences*, **37**(2012), 276–303.
- [32] H. Fukutome, Spin density wave and charge transfer wave in long conjugated molecules, *Progress of Theoretical Physics*, **40**(1968), 1227–1245.
- [33] G.K. Walters, T.L. Estle, Paramagnetic resonance of defects introduced near the surface of solids by mechanical damage, *Journal of Applied Physics*, **32**(1961), 1854–1859.

Наноалмазные коллективные электронные состояния и их локализация

Иван А. Денисов
Андрей А. Зимин
Лесли А. Бурсилл
Петр И. Белобров

Коллективные электронные состояния, несмотря на их важность, редко используются для описания электронной структуры диэлектриков. Природа не связанных с примесями неспаренных спинов, наблюдаемых экспериментально в наноалмазе, всё ещё обсуждается. Мы предлагаем описание их природы в терминах коллективных электронных состояний. Коллективные состояния исследованы точным решением одночастичного одномерного уравнения Шрёдингера в потенциале Кронига – Пенни и первопринципными расчетами волновых функций основных состояний для алмазоидов $\text{C}_{78}\text{H}_{64}$, $\text{C}_{123}\text{H}_{100}$ и $\text{C}_{211}\text{H}_{140}$ методом DFT R-B3LYP/6-31G(d,p). Найдены три различных класса: коллективные связывающие орбитали, соответствующие модулированным решениям для частицы в потенциальной яме; поверхностные несвязывающие проводящие состояния Тамма и подповерхностные связывающие состояния для неоднородно сжатого наноалмаза. Существование неспаренных спинов предположительно объясняется флуктуациями спиновой плотности, значительными для наноразмерных коллективных и протяженных подповерхностных состояний.

Ключевые слова: наноалмаз, наномасштаб, коллективные электронные состояния, состояния Тамма, подповерхностные электронные состояния, парамагнетизм наноалмаза, флуктуация спиновой плотности.